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A Tale of Two Double Layers: Metallic vs. Semiconductor Electrodes

by Payal Chaudhary and Vitaly Alexandrov

Although the electrical double layer (EDL) is central to electrochemistry, its behavior is far from universal. The transition from metallic to semiconductor electrodes introduces complexities that challenge conventional models and demand new theoretical approaches. This perspective article traces the historical trajectory of these two distinct interfaces, contrasting the electronic and

ionic distributions that define their behavior. By examining recent breakthroughs in modeling these disparate systems, we provide a critical outlook on how a deeper understanding of the metal-vs-semiconductor dichotomy can propel the design of more efficient electrochemical energy systems.

The electrical double layer (EDL) at electrode/electrolyte interfaces is central to electrochemical processes, from energy storage and conversion to sensing and corrosion. The classical Gouy-Chapman-Stern (GCS) framework, primarily developed for metallic electrodes, evolved from Helmholtz's parallel-plate capacitor (1879), through the Gouy-Chapman diffuse layer (1910–1913), to Stern's model of specific ion adsorption (1924). Later refinements by Grahame (1947), Bockris and Devanathan (1954), and others introduced more details such as the concepts of the inner and outer Helmholtz planes.^{1–4} Although the exceptional reproducibility of dropping-mercury electrodes established the GCS model as an electrochemical cornerstone, traditional mean-field theories fail to capture many critical atomic-scale details. Also, fundamental disparities between metallic and semiconductor interfaces in electronic structure and charge screening lead to distinct potential distributions, capacitive behaviors, and interfacial reactivity.

In metallic electrodes, electrons are highly delocalized with pronounced density of states near the Fermi level. Charge accumulation upon electrode polarization is screened efficiently over a very short distance (a few atomic layers) via the Thomas-Fermi mechanism. This arrangement results in a thin screening region where excess charge resides primarily on the surface atoms, with electron density “spill-out” into the electrolyte side. The EDL is dominated by the electrolyte response: a compact Helmholtz layer of oriented solvent molecules and specifically adsorbed ions, followed by a diffuse Gouy-Chapman layer in dilute solutions (Fig. 1). This

results in high interfacial capacitance (typically 20–50 $\mu\text{F}/\text{cm}^2$) that is largely governed by the liquid-side structure rather than the internal carrier density or the electronic properties of the bulk metal.

Semiconductor electrochemistry emerged later, spurred by the first commercial silicon transistor developed at Bell Telephone Laboratories in 1954, arguably the most important invention of the 20th century.⁵ The initial attention to silicon and germanium was driven by the need for materials that could operate at the gigahertz frequencies required for radar receivers. When silicon and germanium wafers were etched and cleaned in aqueous electrolytes, they exhibited a range of phenomena inexplicable by the GCS model, such as strong rectification in the dark, capacitance varying by orders of magnitude with applied potential, and light-sensitive currents. Starting in the late 1950s, pioneering work by Brattain, Bardeen, Garrett, Dewald, Pleskov, Davydov, Gerischer, Lewerenz, and others established the foundations of modern semiconductor electrochemistry.^{5–7} Although the first observation of a photoelectrochemical system was made as early as in 1839 by Bequerel, it was Fujishima and Honda who discovered TiO_2 photoactivity in 1972, marking the beginning of modern photoelectrochemistry.⁸ Heinz Gerischer was the first to formulate a comprehensive continuum theory for semiconductor/electrolyte interfaces.^{9–11} In one of his review articles, Gerischer wrote, “The study and kinetic interpretation of electrode reactions was, from the early years of electrochemistry, dominated by processes occurring at electrodes with metallic conductivity. Only since the late 1950s and early 1960s has it been realized that semiconductor electrodes behave differently in many respects...”¹¹ It is somewhat ironic that Gerischer began his groundbreaking investigations of semiconductor electrodes around 1957 at the Max Planck Institute for Metal Research in Stuttgart.

In contrast to metals, semiconductor electrodes exhibit lower carrier densities and a band gap. Charge screening is far less efficient, leading to penetration of the electrical field into the electrode bulk over much larger distances (~ 10 – 1000 nm), forming a space-charge region (Fig. 2). Depending on the applied potential relative to the flat-band condition (conceptually similar to the potential of zero charge for metals), this space charge manifests as band bending: upward (depletion of majority carriers) or downward (accumulation of majority carriers). The space-charge capacitance in series with the Helmholtz and diffuse layers often dominates the total interfacial capacitance, making it highly potential-dependent and sensitive to doping levels, surface states, and defects. For photoelectrochemical applications, band bending drives charge separation, but it also introduces further complexities like recombination of charge carriers at interfacial states.

The limitations of continuum EDL models are increasingly evident in modern electrochemical energy applications, where the “average” behavior described by mean-field theory fails to capture

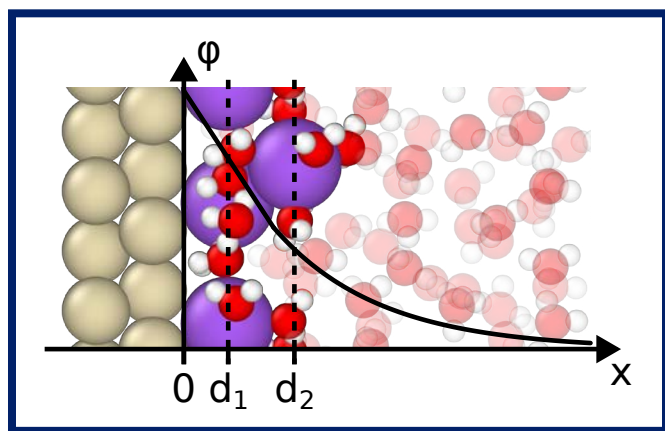


Fig. 1. Structure and potential profile of the electrical double layer (EDL) at a metal/electrolyte interface. Specifically adsorbed ions and oriented water molecules define the inner Helmholtz plane (IHP, $x = d_1$), while the outer Helmholtz plane (OHP, $x = d_2$) marks the closest approach of solvated ions. Beyond d_2 , the potential decays exponentially through the diffuse layer toward the bulk electrolyte value.

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the atomistic and quantum nuances essential for optimizing device performance. Phenomena such as specific ion adsorption, atomic structure of the interface, solvent fluctuations, and quantum effects require microscopic details that continuum theories cannot provide. As we move toward precise interfacial engineering, detailed predictive atomistic and quantum models of the EDL have become indispensable. In this regard, density functional theory (DFT) simulations offer a way to build a deeper fundamental understanding of electronic redistribution at the interface, explicit solvent-ion interactions, and potential-dependent reactivity. Furthermore, the rapid integration of machine learning (ML) force fields is currently transforming the field—extending the accuracy of DFT to the larger time and length scales necessary to capture the dynamic, collective behavior of the EDL in realistic environments.

EDL at Metallic Interfaces: Beyond the Mean-Field Models

The EDL at metallic electrodes has remained a cornerstone of electrochemistry for over a century, yet the traditional model is currently undergoing a fundamental reevaluation. Recent experimental and computational advances have shifted our understanding of electrochemical interfaces from a passive, static region to a grand canonical, quantum, and dynamically fluctuating entity.

From an atomistic perspective, the emphasis has been shifting toward a better understanding of how specific interfacial structures physically shape the double layer's response. For instance, the local bonding environment of reaction intermediates and how they interact with electrolytic cations and anions at the interface has become a significant focus in electrocatalytic research.^{12–14} The local atomic structure of the electrode can also sometimes

affect the EDL behavior more strongly than the electrode inherent chemistry.^{15,16} While classical models assume a uniform surface, atomistic simulations reveal that the specific coordination of atoms profoundly alters the local electric field and solvent orientation. Thus, computational models of interfacial reactivity move beyond simple descriptors such as adsorption energies of reaction intermediates to how they dynamically evolve within the electrode/electrolyte microenvironment.

Although much of electrocatalytic theory has historically been enthalpy-driven, an emerging frontier focuses on the entropy of the EDL as an important contributor to the overall reaction energetics. Drawing parallels with biomolecular systems where conformational entropy often governs substrate binding, electrochemists have started quantifying how interfacial solvent disorder modulates the Gibbs free energy of reactions ($G = H - TS$). For example, the potential of zero charge (PZC) is now understood not just as a static electronic property of the metal, but as a parameter highly sensitive to the configurational entropy of the interfacial solvent.¹⁷ This relationship is experimentally accessible via the potential of maximum entropy (PME), typically characterized using laser-induced current transient (LICT) techniques or through *ab initio* molecular dynamics (AIMD). A highly structured, “ice-like” water layer at the PZC represents an entropic penalty that must be overcome during the solvent reorganization required for interfacial charge transfer. Peterson et al. recently demonstrated that under certain electrolyte compositions multiple stable configurations can exist, each corresponding to a distinct PME value, with one coinciding with the PZC.¹⁸

Another critical development in computational electrochemistry is the transition from constant-charge to constant-potential simulations. Historically, DFT was constrained to a fixed number of electrons, a limitation that caused the electrode potential to drift as interfacial ions reorganized. Modern grand-canonical DFT (GC-DFT) solves this by allowing electrons to flow continuously in and out of the electrode to maintain a constant Fermi level.^{19–21} This breakthrough

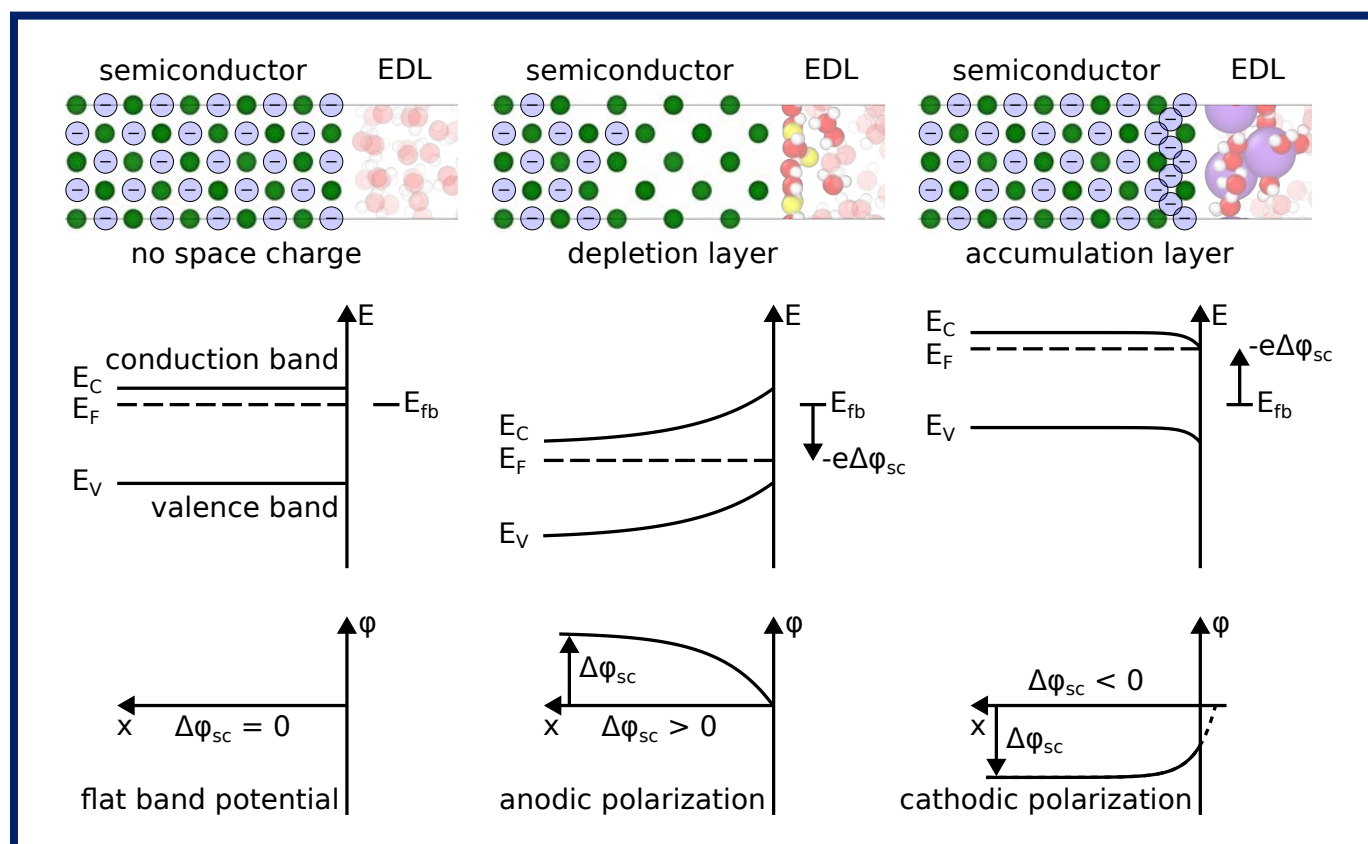


Fig. 2. Space charge layers at an n-type semiconductor/electrolyte interface under three bias conditions. Top: charge carrier distributions; middle: corresponding band edge diagrams showing the conduction band (E_C), valence band (E_V), and Fermi level (E_F) relative to the flat-band potential (E_{fb}); bottom: potential profiles across the interface. Left to right: flat-band condition ($\Delta\phi_{sc} = 0$), anodic polarization forming a depletion layer ($\Delta\phi_{sc} > 0$), and cathodic polarization forming an accumulation layer ($\Delta\phi_{sc} < 0$).

enables the evaluation of capacitance and reaction barriers as explicit functions of applied voltage, providing a direct bridge to experimental potentiostatic measurements.

In addition to atomistic details for both the electrode and electrolyte, there is a growing interest in quantum features of electrochemical interfaces. The metal surface is no longer viewed as a “hard wall.” Instead, the electronic density spills out into the solvent, creating a quantum-mechanical tail. This electronic overlap generates a non-monotonic relationship between the PZC and local permittivity, explaining the anomalous capacitance peaks observed in metals like gold and silver—a phenomenon that directly contradicts the monotonic predictions of classical Gouy-Chapman-Stern theory.²²

The quantum mechanical aspects of the EDL extend beyond the electronic spill-out of the metal to the very nuclei of electrolyte species. In the aqueous environments that dominate modern electrochemistry, the behavior of hydrogen—the smallest and most ubiquitous ion—defies classical description.^{23,24} Path-integral molecular dynamics (PIMD) have fundamentally demonstrated that interfacial protons behave less like localized point particles and more like delocalized waves predicting enhanced probability of water dissociation at the aqueous Pt interfaces.²⁵ These nuclear quantum effects (NQE) are far from mere computational nuances; they can strongly affect solvent dynamics and interfacial kinetics.^{26–28} By accounting for the wave-like nature of the proton, researchers have found that energy barriers for proton transfer are significantly lowered via tunneling, while the zero-point energy (ZPE) corrections can entirely reorder the stability of adsorption sites. On benchmark surfaces like Pt(111), NQEs shift the preferred site for hydrogen adsorption, a factor that is indispensable for accurately predicting the kinetics of the hydrogen evolution reaction (HER).

The implications of NQEs extend far beyond HER.^{29,30} Because most complex electrochemical transformations such as CO₂ and NO₃ reduction reactions rely on coupled proton-electron transfers, the quantum nature of both electrons and protons can play an important role. For instance, the branching ratio between methane and ethylene often hinges on the precise timing and energetics of a hydrogenation step. If a proton can tunnel through a barrier to reach an intermediate more easily than it can participate in coupling, the entire product distribution shifts. In fact, recent work used electron tunneling at metallic interfaces to study electrocatalytic reactions.³¹ Overall, ignoring quantum effects may lead to an overestimation of activation barriers and a fundamental misunderstanding of how local pH and electrolyte composition modulate catalyst selectivity.

EDL at Semiconductor Interfaces: Coupling Dynamics and Chemistry

Historically, the EDL at semiconductor/liquid interfaces has remained less understood than its metallic counterparts, largely due to the inherent complexity of potential distribution and charge carrier behavior within the solid. However, the push for efficient solar-to-chemical energy conversion and the proliferation of semiconductor-based technologies have catalyzed a significant shift in our fundamental understanding. Much like the evolution of metal/liquid interface research, the emphasis is now transitioning toward bridging the gap between quantum-mechanical precision and large-scale, long-time dynamics. This leap is being driven by the integration of grand canonical density functional theory (GC-DFT) and emerging ML tools. Here, we emphasize the dual-layer nature of the semiconductor interface: while metals only have an electrolyte-side double layer, semiconductors feature a simultaneous space-charge layer within the solid itself (see Fig. 2).⁷

The structural organization of the EDL at semiconductor interfaces differs fundamentally from that of metals due to the lower carrier density of the solid.^{11,32} In metals, a high density of electronic states enables near-perfect charge screening, concentrating the potential drop into a compact, high-capacitance Helmholtz layer just a few Ångströms thick. This “overscreening” often creates extreme interfacial field strengths and a rigid layer of hydrated ions. In contrast, the potential at a semiconductor interface is partitioned between the external EDL and an internal space-charge region. Because semiconductors exhibit

distributed screening, the total interfacial capacitance is typically dominated by internal Mott-Schottky behavior, meaning energy storage and electrical field profiles are governed by internal doping rather than external ion crowding.

This partitioning creates a unique electrochemical microenvironment.^{7,33} By limiting the potential drop across the liquid-side Helmholtz layer, semiconductors avoid the chaotic ion restructuring and extreme field strengths characteristic of metals. This leads to two major advantages. First, the energy specificity of semiconductor bands allows for superior product selectivity. While metals drive multiple competing reactions via an unmitigated surface potential drop, semiconductor band edges can be “pinned” to align specifically with desired redox energy levels. Second, semiconductors promote local pH stability. The more controlled charge transfer and lower current flux at these interfaces prevent the drastic surface pH deviations common at metals, allowing for precise electrolyte tuning and higher faradaic efficiencies.

As mentioned earlier, grand-canonical DFT (GC-DFT) enables constant-potential instead of standard fixed-charge DFT simulations of electrochemical interfaces. GC-DFT allows one to model potential partitioning at the interface accurately, showing how much of the voltage drop occurs across the Helmholtz layer versus the space-charge region. This distinction is vital for photoelectrocatalysis, where the driving force for carrier extraction depends entirely on this internal potential distribution. However, semiconductors represent a more challenging case than metals from the electronic-structure point of view since proper treatment of the band gap and charge carrier localization requires approaches beyond standard DFT, such as hybrid exchange-correlation functionals or the GW approximation.³⁴ This requirement typically adds much higher computational costs. Combined with the need to monitor the dynamics of charge carriers such as electron holes or polarons generated by sunlight, it adds enormous computational burden to ab initio calculations of semiconductor/liquid interfaces. Another complication may arise in highly doped semiconductors or thin films, when the contraction of the space-charge region would allow quantum confinement and tunneling effects. This effect can substantially reduce the EDL capacitance by introducing a quantum capacitance in series with the Helmholtz capacitance. These effects are particularly significant in nanoscale devices.

Traditionally, the semiconductor space-charge region and the electrolyte EDL were modeled as two independent regions. Modern research recognizes, however, that these regions are dynamically entangled: the potential distribution across the interface shifts continuously during a reaction. Moving beyond the static framework toward an atomistic understanding of the coupling between semiconductor charge carriers and redox-active species faces significant hurdles. One challenge is the timescale mismatch: electron-hole generation and recombination occur on picosecond to nanosecond scales, whereas ion reorganization in the electrolyte is orders of magnitude slower. Despite these complexities, recent advances in ML force fields hold great promise for accurately modeling such complex, dynamic interfaces.

Machine-Learning Assisted Studies of the EDL

Recent years have seen transformative advances in ML tools for electrochemical systems. The introduction of ML interatomic potentials (MLIPs) trained on DFT data—including energies, forces, and virial stresses—enables spatially and temporally large-scale dynamical simulations with near-DFT accuracy (see Fig. 3).³⁵ Frameworks such as DeePMD-kit³⁶ and NequIP³⁷ have successfully simulated aqueous interfaces with thousands of atoms, uncovering complex hydrogen-bonding networks and proton-transfer mechanisms that are inaccessible via traditional ab initio molecular dynamics (AIMD).

An emerging frontier is the integration of grand-canonical DFT (GC-DFT) with MLIPs to simulate interfaces at a constant

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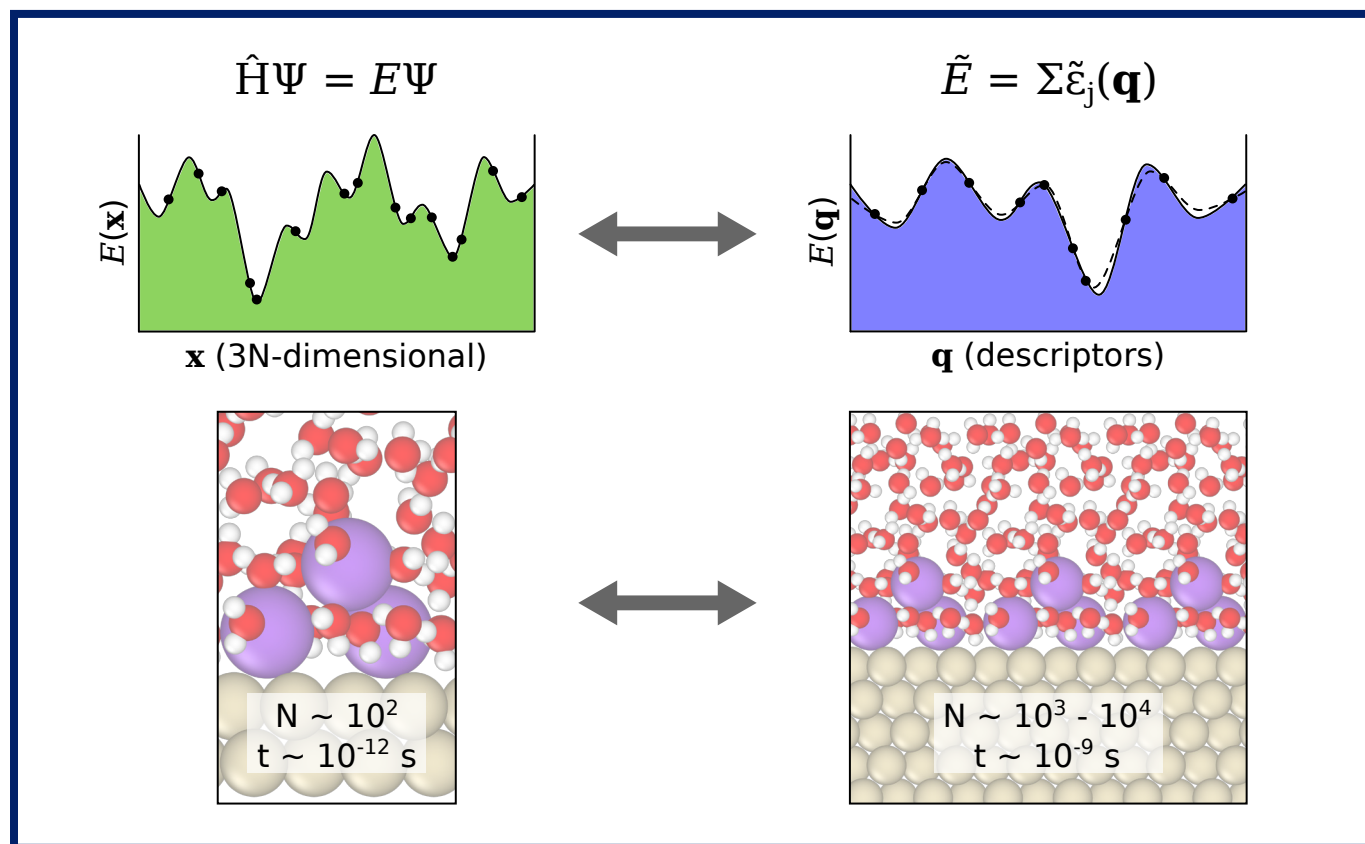


Fig. 3. Comparison of *ab initio* and machine learning interatomic potential (MLIP) approaches to molecular dynamics simulation of the electrode/electrolyte interface. Left: the exact potential energy surface $E(x)$ defined via the Schrödinger equation, limited to small system sizes ($N \sim 10^2$) and short timescales ($t \sim 10^{-12}$ s). Right: MLIPs approximate the total energy as a sum of environment-dependent atomic contributions, mapping atomic configurations onto low-dimensional descriptors q . The dashed line indicates close agreement between the ML-predicted and reference energy surfaces, enabling simulations of $N \sim 10^3$ - 10^4 atoms over timescales reaching $t \sim 10^9$ s.

electrochemical potential. This synergy overcomes the intrinsic limitations of traditional GC-DFT, which is typically restricted to small systems and short timescales. Recent architectures, such as Graph Neural Networks (GNNs), have begun to directly incorporate electrode potential as an input to capture field-dependent dynamics.^{38,39} However, while most studies focus on metallic interfaces, semiconductor systems present distinct challenges and opportunities owing to the need to accurately describe band gaps and localized electron/hole polarons.

Describing charge carrier dynamics in semiconductors requires moving beyond standard MLIPs, which typically map atomic positions to a single potential energy surface (PES). Emerging hybrid frameworks offer a promising path forward. For example, Leopold (LEarning Of POLaron Dynamics) incorporates the charge state as an input feature alongside atomic coordinates, enabling nanosecond-scale simulations of polaron hopping at DFT-level accuracy.⁴⁰ This ability allows the model to distinguish between multiple PESs for a single configuration, such as a polaron localized on one lattice site versus another. Similarly, DeepWannier models track electronic centers of mass (Wannier centroids), providing a rigorous bridge between atomic motion and carrier dynamics.³⁶ Despite these breakthroughs, applying such hybrid ML approaches to the semiconductor/liquid EDL remains rare, representing an open frontier for theoretical research.

Looking forward, the shift from mean-field approximations to a fully atomistic, quantum-mechanical framework marks a defining moment for interfacial electrochemistry. The evolution of EDL models from the rigid parallel plates of Helmholtz to the grand-canonical, quantum-level simulations of today reflects the deepening integration of solid-state physics, surface chemistry, and data science.

Merging predictive, quantum-grounded modeling with atomic-scale engineering will be instrumental in optimizing the performance of next-generation electrochemical systems. Over the coming decade, resolving the “modeling gap” will hinge on integrating grand-canonical DFT with ML tools, facilitating high-fidelity simulations of field-dependent dynamics and reactivity across nanosecond timescales.

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References

1. C. M. Schott, P. M. Schneider, K. T. Song, H. Yu, R. Götz, et al., *Chem Rev*, **124**, 12391 (2024).
2. J. Wu, *Chem Rev*, **122**, 10821 (2022).
3. W. Schmickler, *Chem Rev*, **96**, 3177 (1996).
4. W. Schmickler and D. Henderson, *Prog Surf Sci*, **22**, 323 (1986).
5. M. Riordan, L. Hoddeson, and C. Herring, *Rev Mod Phys*, **71**, S336 (1999).
6. M. G. Walter, E. L. Warren, J. R. McKone, S. W. Boettcher, Q. Mi, et al., *Chem Rev*, **110**, 6446 (2010).
7. R. Memming, *Semiconductor Electrochemistry*, (Wiley, 2015).
8. A. Fujishima and K. Honda, *Nature*, **238**, 37 (1972).
9. H. Gerischer, *Surf Sci*, **18**, 97 (1969).
10. H. Genscher, *J Electrochem Soc*, **113**, 1174 (1966).
11. H. Gerischer, *Electrochim Acta*, **35**, 1677 (1990).
12. M. C. O. Monteiro, F. Dattila, B. Hagedoorn, R. García-Muelas, N. López, et al., *Nat Catal*, **4**, 654 (2021).
13. P. Li, Y. Jiao, J. Huang, and S. Chen, *JACS Au*, **3**, 2640 (2023).
14. J. Resasco, *JACS Au*, **5**, 5253 (2025).
15. S. Xue, P. Chaudhary, M. R. Nouri, E. Gubanova, B. Garlyyev, et al., *J Am Chem Soc*, **146**, 3883 (2024).
16. N. L. Fröhlich, J. Liu, K. Ojha, A. Hagopian, K. Doblhoff-Dier, et al., *Nat Chem* (2026).
17. X. Ding, B. Garlyyev, S. A. Watzel, T. Kobina Sarpey, and A. S. Bandarenka, *Chem Eur J*, **27**, 10016 (2021).
18. A. S. Petersen, T. K. Madsen, T. K. Sarpey, C. M. Schott, E. L. Gubanova, et al., *Adv Theory Simul*, **9**, e00958 (2026).
19. A. J. W. Wong, D. Zhu, S. Chatterjee, and M. J. Janik, *J Phys Chem C*, **129**, 20857 (2025).
20. N. G. Hörmann, O. Andreussi, and N. Marzari, *J Chem Phys*, **150**, 041730 (2019).
21. R. Sundararaman, W. A. Goddard, and T. A. Arias, *J Chem Phys*, **146**, 114104 (2017).
22. W. Schmickler, *Chem Rev*, **96**, 3177 (1996).
23. J. A. Morrone, and R. Car, *Phys Rev Lett*, **101**, 017801 (2008).
24. M. Ceriotti, J. Cuny, M. Parrinello, and D. E. Manolopoulos, *Proc Natl Acad Sci USA*, **110**, 15591 (2013).
25. J. Lan, V. V. Rybkin, and M. Iannuzzi, *J Phys Chem Lett*, **11**, 3724 (2020).
26. T. E. Markland, and M. Ceriotti, *Nat Rev Chem*, **2**, 0109 (2018).
27. N. Stolte, J. Daru, H. Forbert, J. Behler, and D. Marx, *J Phys Chem Lett*, **15**, 12144 (2024).
28. M. Sun, B. Jin, X. Yang, and S. Xu, *Nat Commun* **2025** *16:1*, **16**, 3600 (2025).
29. T. E. Markland and M. Ceriotti, *Nat Rev Chem* **2018** *2:3*, **2**, 0109 (2018).
30. M. Ceriotti, W. Fang, P. G. Kusalik, R. H. McKenzie, A. Michaelides, et al., *Chem Rev*, **116**, 7529 (2016).
31. J. H. K. Pfisterer, Y. Liang, O. Schneider, and A. S. Bandarenka, *Nature*, **549**, 74 (2017).
32. C. P. Ursenbach and G. A. Voth, *J Chem Phys*, **103**, 7569 (1995).
33. A. J. Bard, A. B. Bocarsly, F. R. F. Fan, E. G. Walton, and M. S. Wrighton, *J Am Chem Soc*, **102**, 3671 (1980).
34. J. M. Crowley, J. Tahir-Kheli, and W. A. Goddard, *J Phys Chem Lett*, **7**, 1198 (2016).
35. M. Kulichenko, B. Nebgen, N. Lubbers, J. S. Smith, K. Barros, et al., *Chem Rev*, **124**, 13681 (2024).
36. J. Zeng, D. Zhang, A. Peng, X. Zhang, S. He, et al., *J Chem Theory Comput*, **21**, 4375 (2025).
37. S. Batzner, A. Musaelian, L. Sun, M. Geiger, J. P. Mailoa, et al., *Nat Commun*, **13**, 2453 (2021).
38. J. L. Chen, X. Z. Qi, J. Zhu, J. Li, X. C. Jiang, et al., *J Chem Theory Comput*, **21**, 11039 (2025).
39. R. Wang, S. Fang, Q. Huang, and Y. Liu, *J Chem Theory Comput*, **21**, 7628 (2025).
40. V. C. Birschtzky, L. Leoni, M. Reticioli, and C. Franchini, *Phys Rev Lett*, **134**, 216301 (2025).